

Tuning Some Optical Parameters of SeA_x (A= Fe:Sn) Chalcogenides For photovoltaic Applications.

¹K. A. Aduloju, ²Nicholas Saidu, ³A.T Fatigun, ⁴O.R Salau, ⁵G. E. Adesakin,
⁶Fatai Ayodele, ⁷O. O. Olusola.

¹Department of Physics, Faculty of Science, Ekiti State University, P.M.B. 5363, Ado-Ekiti, , Nigeria.

² Department of Physics/mathematics Education, Kogi State College of Education, Kabba, Nigeria.

^{3,4,5,6}Department of Physics, Faculty of Science, Ekiti State University, P.M.B. 5363, Ado -Ekiti, , Nigeria

⁷ Department of Physics, School of Science, College of Education, IkereEkiti, Ekiti State, P.M.B. 250, Nigeria.

Corresponding Author: K. A. Aduloju

Abstract ; Polycrystalline SeFeSn thin films have been fabricated with different molar concentrations of Iron :Selenium (Fe:Se) using chemical spray pyrolysis at 400K, Spectroscopic measurements within the UV-Vis-NIR range of the thin film samples were studied at room temperature in order to understand their optical properties. The deposited films are characterized by wide bandgapenergies in the range 3.75-3.87eV , the films bandgapwere observed to be tuned from 3.75eV for Fe:Se(0.05:0.05) molar concentration to 3.87 eV for Fe:Se(0.1: 0) molar concentration in a parabolic-increase trend . These demonstrate promising optical properties of the ternary film-samples for photovoltaic technology and good miscibility of alloyed constituents in the Seleniumhost crystal lattice. The refractive index, dielectric constant and extinction coefficient were evaluated, the extinction coefficient shows a decay in value as the molar concentration of Fe increases with a simultaneous decrease in the Sn molar concentration in the multicomponent films.

Keywords; Polycrystalline, molar concentrations, spectroscopic measurements, miscibility, photovoltaic, refractive index , dielectric constant, extinction coefficient .

Date of Submission: 21-01-2018

Date of acceptance: 03-02-2018

I. Introduction

In the last few years, considerable interest has been shown in the synthesis of semiconducting nanocrystalline films for optoelectronic applications. Nanostructured materials based on II-VI group compounds have long been known to be suitable for photovoltaic device applications because of their high optical absorption coefficients [1]. Likewise, Ternary (groupV-VI) compounds of type A_2B_3 (A = Bi and B = S, Se or Te) are considered important semiconductor materials due to their low cost fabrication techniques and abundance of the constituent elements [2].

However, the main advantage of binary and ternary chalcogenide semiconductors is seen in their promise of lower cost; since less energy for materials are relatively lower in cost and large scale productions are feasible [3] . Consequently, the synthesis of ternary metal chalcogenidegroups semiconductors in nanocrystalline form has been a rapidly growing area of research due to their important non-linear optical properties, luminescent properties, quantum confinement effect and other important chemical and physical properties [1].

Moreover, Semiconductors based on selenium are important class of semiconducting systems which have been widely studied too due to their fundamental electronics and optical properties. Intensive researches have been performed in the past to study the fabrication and characterization of these compounds in the form of films [4]. Examples are PbSe [5,6], SnSe [4,7], CdSe [8,9], AgSe [10], ZnSe [11] and SbSe [12]. Furthermore, Several researchers have investigated the optical and structured properties of Iron Selenide (FeSe) [4,7,13]. Literatures equally revealed some binary compounds with selenium as one of their components such as; Bismuth Selenide (Bi_2Se_3) and Silver Selenide (Ag_2Se) as narrow bandgap semiconductors with n-type conductor which are applicable in; switching devices, infra red detectors and thermoelectric generators [14].

Also, the pseudo-binary alloy $\text{Fe}_{0.03}\text{Sn}_{0.97}$ has been found to be a p-type semiconductor which could be used as a suitable material for the fabrication of thermoelectric devices [15].

In this study , we investigate the optical properties of thin-film of ternary compound formed from a combination Tin,Iron and Selenide (FeSnSe) prepared by spray pyrolysis with the hope of developing a material that could act as a substitute to the conventional silicon-based solar cell. Eventhough the combination contained earth abundant elements with high absorption coefficient and good stabilities but there have been no reports in the literature for the optical and electrical properties of the multicomponent thin film . Optical

parameters such as absorbance, transmittance, reflectance, absorption coefficient, energy band gap, optical conductivity and dielectric constants of FeSnSe would be revealed in our studies.

II. Materials and Methods

2.1 Substrate Preparation

Owing to rigidity, hardness, chemical inertness, flat and smooth surface with good transmission characteristics of the substrate needed for this investigation, borosilicate glass was used. The substrates were first washed with detergent and distilled water to remove contaminant and glass-stains, then ultrasonically cleaned for ten minutes with isopropyl alcohol. Dried and kept in desiccator for a later use.

2.2 Materials Used

The materials used in the deposition of Iron-Tin-Selenide (SnFeSe) by spray pyrolysis are:

- i. Tin chloride ($SnCl_2$)
- ii. Iron chloride ($FeCl_2$), and
- iii. Selenium dioxide (SeO_2)

The precursors used for samples fabrication were prepared following six different compositions, the variation in molar concentration of the elements at each spray is as display on "table 2"

Table 2, The Molar ratio of Fe:Sn (Moles) in Selenium

Samples	Fe	Sn	Se
A	0.00	0.10	0.20
B	0.05	0.05	0.20
C	0.06	0.04	0.20
D	0.07	0.03	0.20
E	0.08	0.02	0.20
F	0.10	0.00	0.20

With the aid of syringe which was well washed, cleaned with distilled water and ethanol, 2ml of methoxyethanol was added to each precursor to slow down the reaction sufficiently.

2.3 Methodology

Spray pyrolysis prove to be a method suitable for the deposition of large area metal oxide, spinel oxide, and different chalcogenide thin films [16,17]. This process is widely applied and very attractive for the deposition of low cost thin film photovoltaic solar cells. The set-up is shown in "Fig 1".

During the spray pyrolysis process, the precursor aerosol was sprayed towards the heated substrate. The component in the precursor droplets react to form film on the substrates surface and some by-products vaporized into the open atmosphere. Since high temperature was involved, the spraying-chamber was kept from moisture to avoid spark. The deposition time was 10 minutes and the deposition temperature was kept at 400°C.

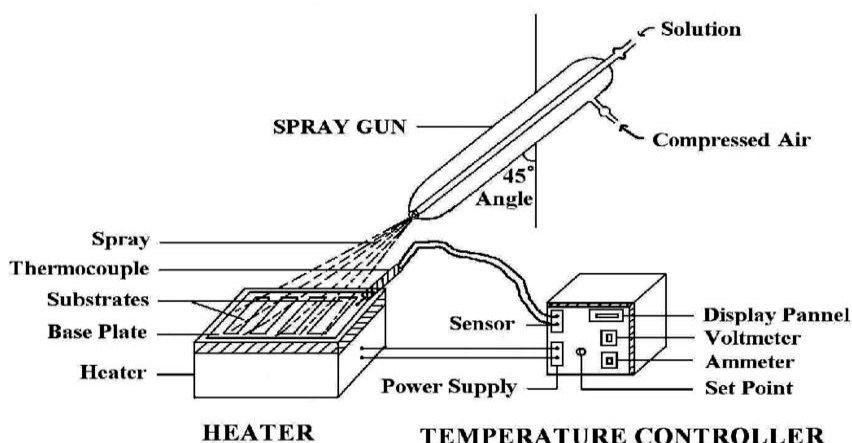
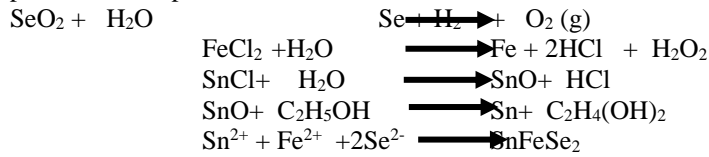


Figure 1, Spray Pyrolysis Deposition Apparatus [18]

Six samples from six different composition of precursors were use separately. After each deposition, the spraying chamber is cleaned before been re-used, the Metal-chalcogenide thin films deposited on glass slide were cooled by air quenching and used as prepared.

The basic chemistry involved in the complex ion formation of the thin-film of SnFeSe during the spraying processes is as presented below;



III. Measurements

In both crystalline and amorphous semiconductors, the absorption coefficient near the fundamental absorption edge is dependent on photon energy. In the high absorption region, the absorption coefficient takes on the following more general form as a function of photon energy ;

$$ahf = A(ahf - E_g)^n \text{ where; } n=1/2 \text{ for direct transitions and } 2.1$$

$$ahf = B(ahf - E_g)^n \text{ Where; } n=2 \text{ for indirect transitions } 2.2$$

where f is the frequency of the incident photon, h is the Planck's constant, A and B are constants, and E_g is the optical energy gap. α is the absorption coefficient given by;

$$\alpha = \frac{\text{Absorbance}}{\text{optical density}} = 2.303(A/t) \quad 2.3$$

where; A is absorbance and t is the film thickness.

The optical transmittance T is related to the absorption coefficient α and Refractive index n by ;

$$T = (1-R)^2 \exp(-\alpha d) / (1-R_2) \exp(-2\alpha d) \quad 2.4$$

The extinction coefficient K is related to α by ;

$$K = \alpha \lambda / 4\pi \quad 2.5$$

The optical conductivity σ_o is given by ;

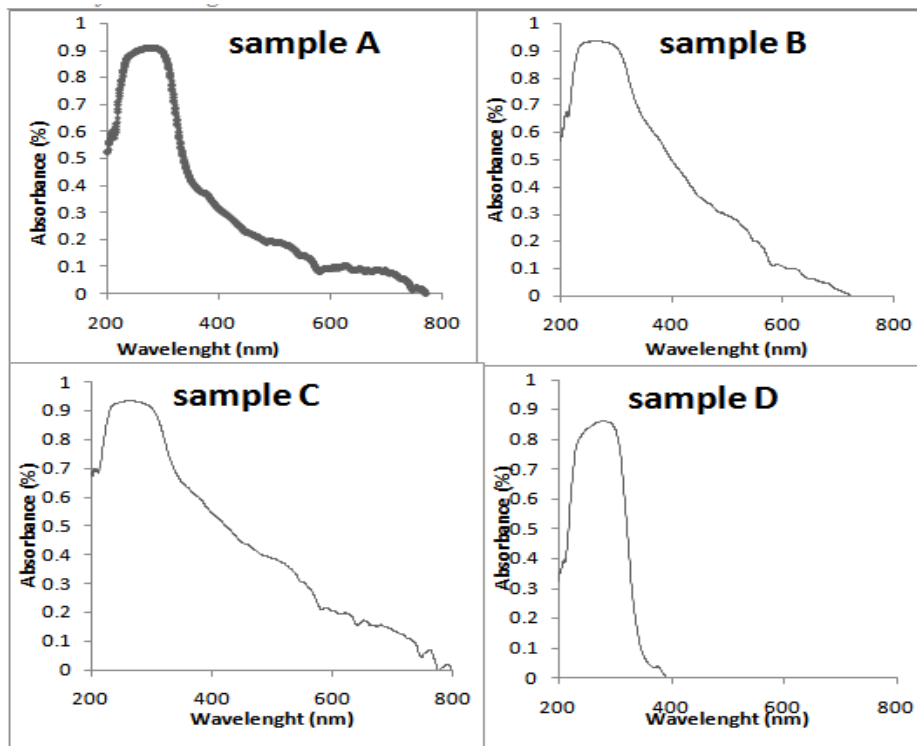
$$\sigma_o = \frac{nc}{4\pi} \quad 2.6 \text{ Where; } c \text{ is the velocity of light, In metals, } \sigma_o \text{ and } k \text{ are very high as reflectance approaches unity.}$$

The dielectric constant ϵ is related to K and n , it is defined as the response of the material towards the incident electromagnetic field. The dielectric constant of a compound is divided into two parts: real and imaginary, and can be written as ;

$$\epsilon^* = \epsilon_r + i\epsilon_i \quad 2.7$$

IV. Result and Discussions

The thicknesses of the deposited films as determined by Profilometer were within the range of 100nm – 105nm. The optical properties such as; absorbance, reflectance etc, of the film-samples were determined by Spectrophotometer within UV-Vis-NIR (200nm to 800nm) range, the spectrophotometer works in conjunction with a computer aided with an Avasoft software.



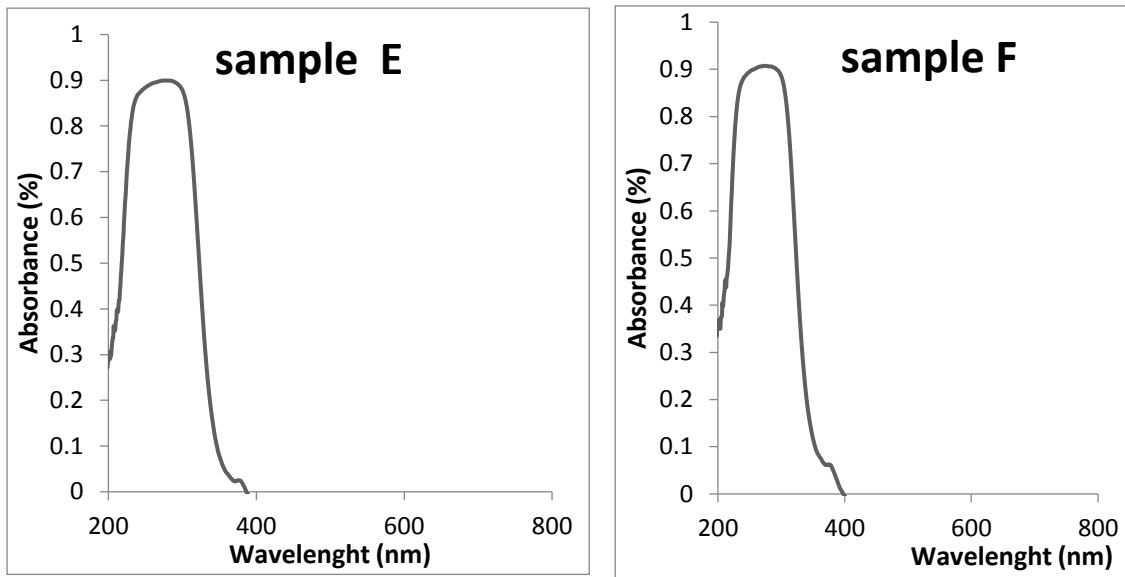
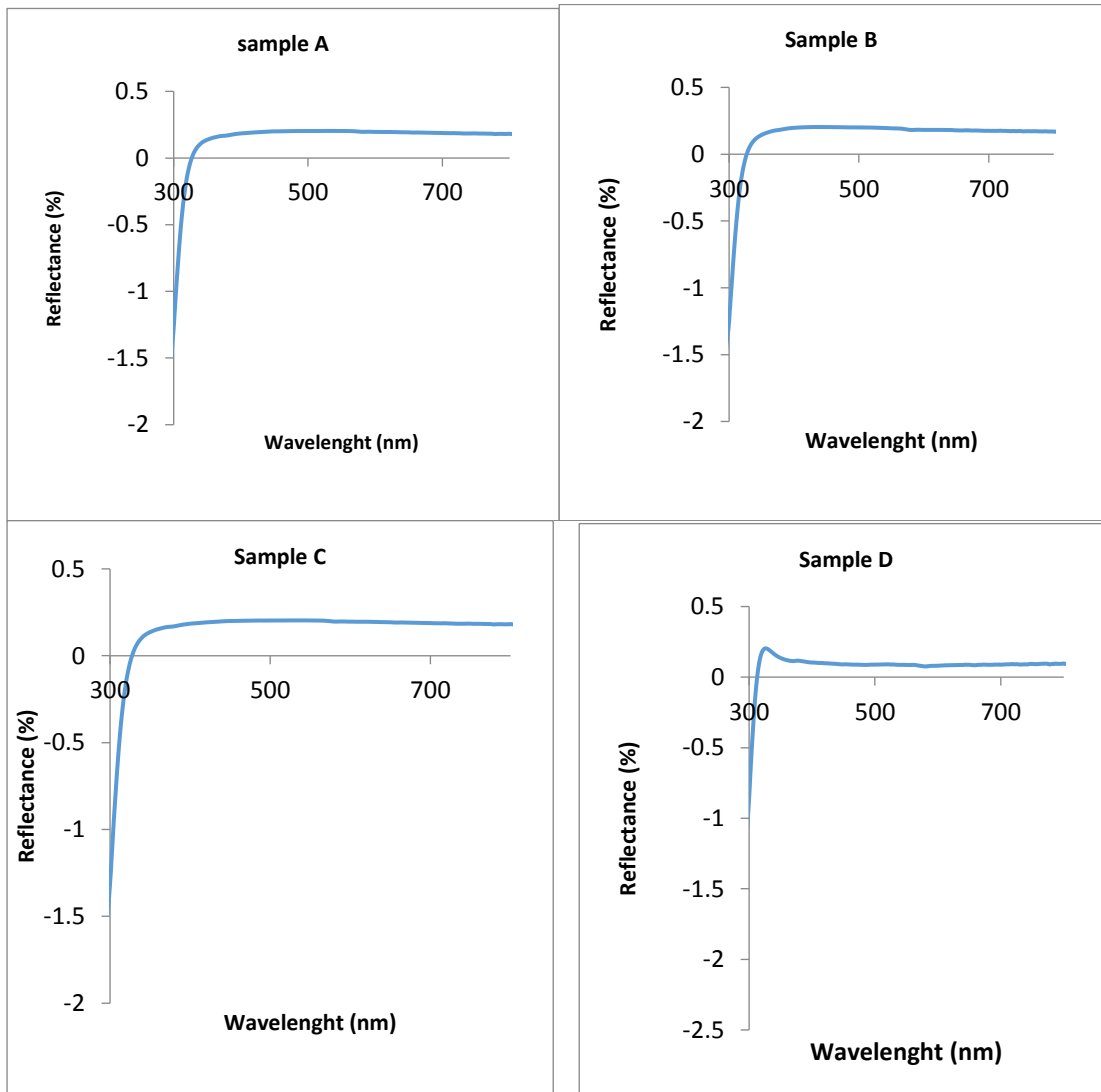


Figure 1, Absorbance Spectra of the different Samples.



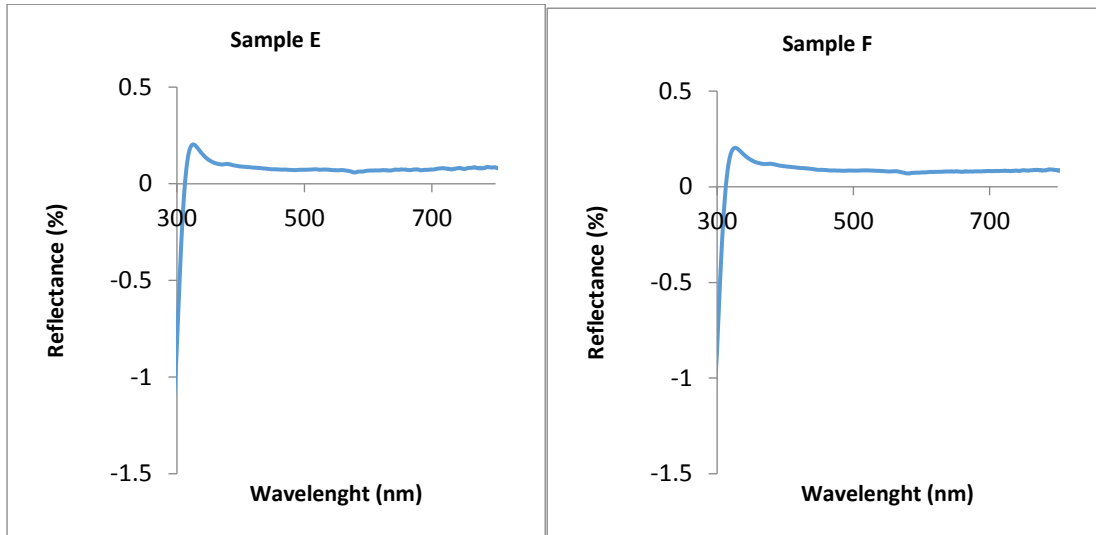
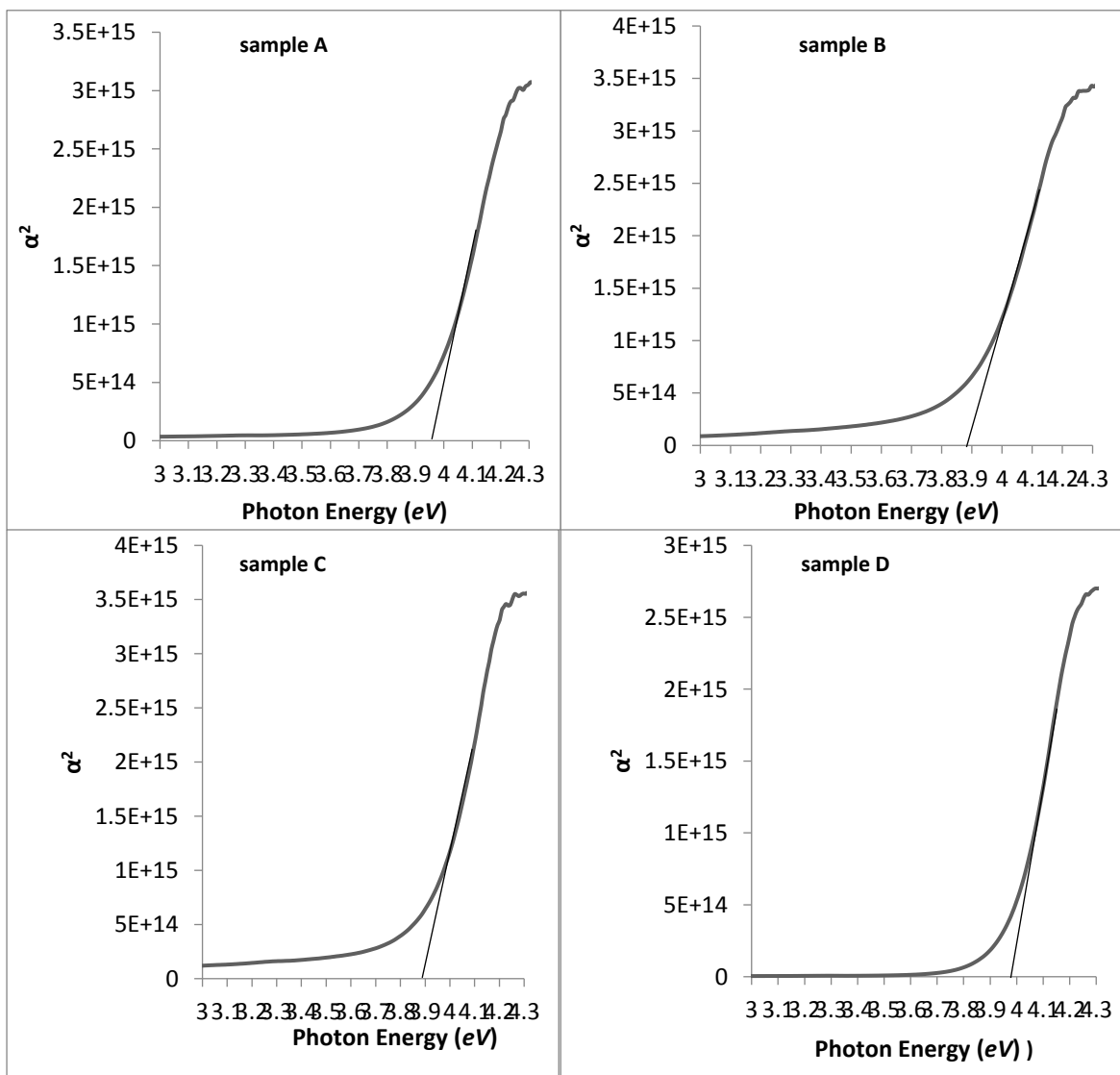


Figure 2, Reflectance Spectra of the different Samples.



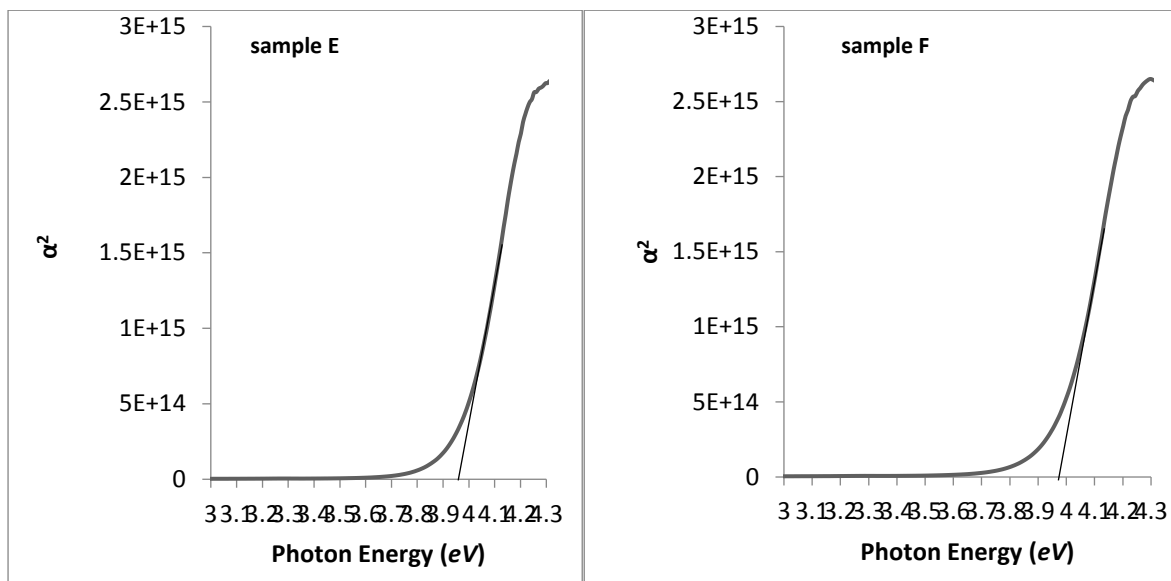


Figure 3, α^2 versus Photon Energy (eV) for the different Samples.

The absorption Spectra for the different Se:Fe concentrations in the samples are shown in ‘‘fig 1’’. From the graph , it is clear that there is an improvement in optical absorption spectra of samples A,B and C within the visible region while samples D , E and F recorded Zero value of absorption in same region. However , all samples examined had improved absorption in the Ultra-violet region . In general, the intensity of absorption peak was recorded in the UV-Vis region followed by a sudden decay in Vis-NIR region, It is evidence that samples D,E and F which recorded virtually Zero absorbance in the Vis – NIR region could be used as transparent conducting material.

‘‘Fig 2’’ shows how the reflection response changes in the different film-samples examined. Within the UV-Vis-NIR range nearly all the samples recorded low reflectance except for a very little improvement for samples A ,B and C, this reflects the information that nearly all the samples could be used as anti-reflector materials.

The plots of α^2 versus Photon Energy ($E=h\nu$) for the different Samples are depicted in ‘‘fig 3’’. The forbidden gap / Energy gap (E_g) is obtained by the extrapolation of the linear portion of the curve to $\alpha^2 = 0$ (where $E=h\nu = E_g$). However, the summary of results obtained from the plots is display on a bar chart in ‘‘fig4’’. It is evident that the E_g recorded for the film-samples falls within the range 3.75 – 3.85eV , the peak value of 3.87eV was recorded in sample F while sample B had the lowest E_g of 3.75eV, also samples A and E are of equal value of E_g . These revealed the fact that all the film samples are wide band-gap material that could be useful in photovoltaic technology.

Other parameters recorded in the course of furthering investigation on the optical properties of the thin-samples are shown in ‘‘fig 5-8’’. The refractive indices obtained are within the range 2.77 -3.60 with sample A having the highest and B with the least . The extinction coefficient takes the value 1.54 - 4.43 , sample C had the highest and E the least. For the optical conductivity (σ_o) of the samples, the highest value of 56.5 was obtained for sample A and the lowest 4.21 for sample F , generally there is a decrease in σ_o as the Molar concentration of Sn increases/Fe decreases . In addition , the dielectric constants recorded were in the range 15.34 -29.92 with sample A having the least and B having the highest.

In summary, the high value of the refractive index equally attested to the usefulness of SeFeSn –ternary films in photovoltaic applications, also the highest value in dielectric permittivity recorded by sample B (where Fe:Sn is 0.5:0.5M) can be attributed to the equal contributions of the multicomponent polarizability, which are deformational and relaxation. Deformational polarizability is the mutual displacement of the oppositely charged particles under the action of applied field. On the other hand, the relaxation polarizability originated from limited mobility of the permanent dipoles usually at lower frequencies. The unequal multicomponent polarizability contributions brought about the low values of ϵ recorded in the other samples.

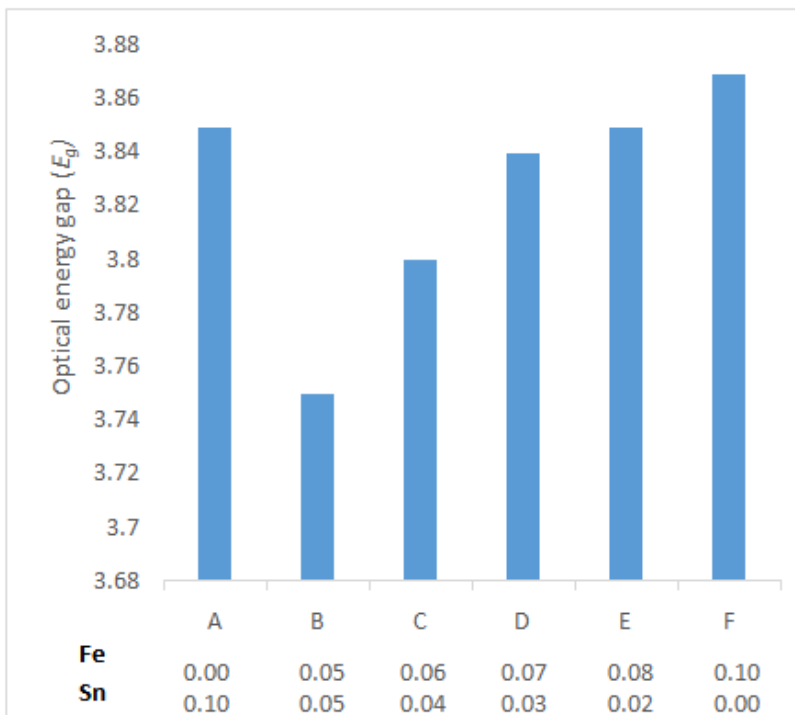


Figure 4, Optical energy gap (E_g) versus sample compositions

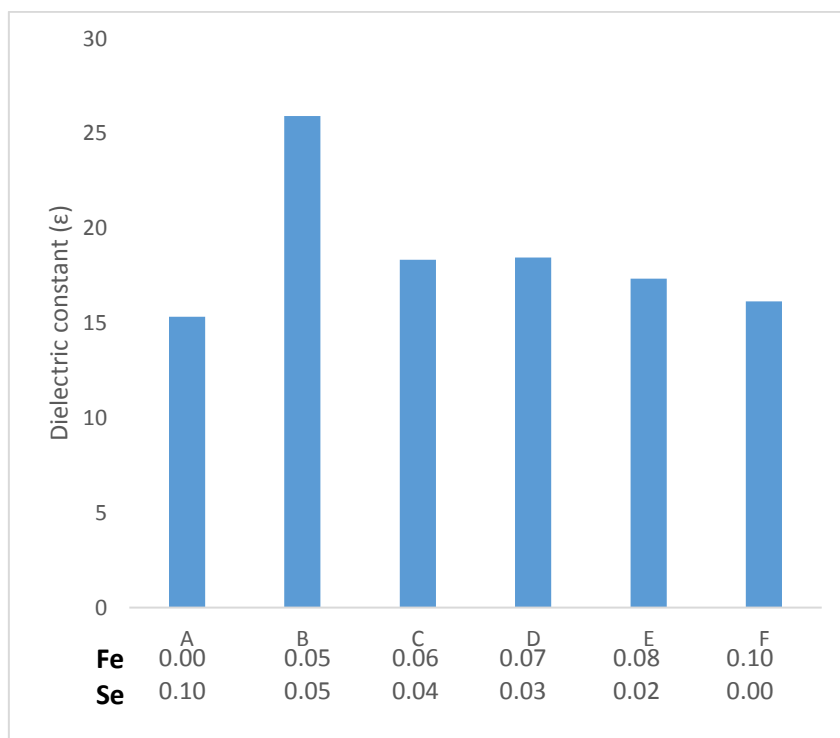


Figure 5, Dielectric constant (ϵ) versus sample compositions

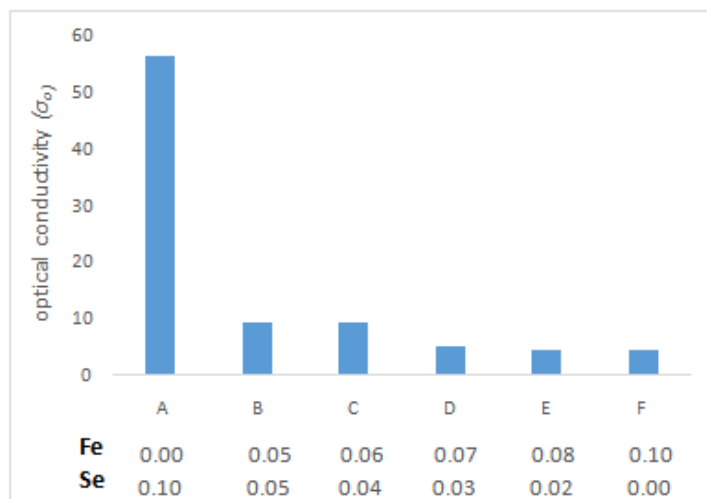


Figure 6, Optical Conductivity (σ_o) versus sample compositions

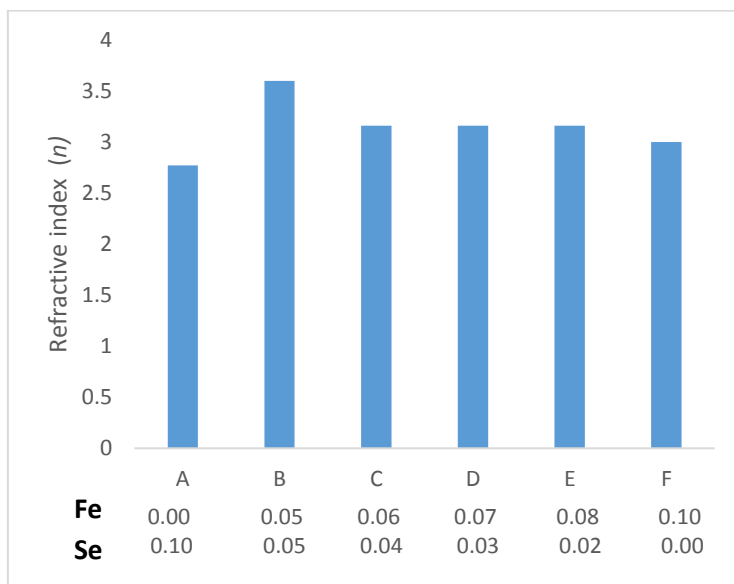


Figure 7, Refractive index (n) versus sample compositions

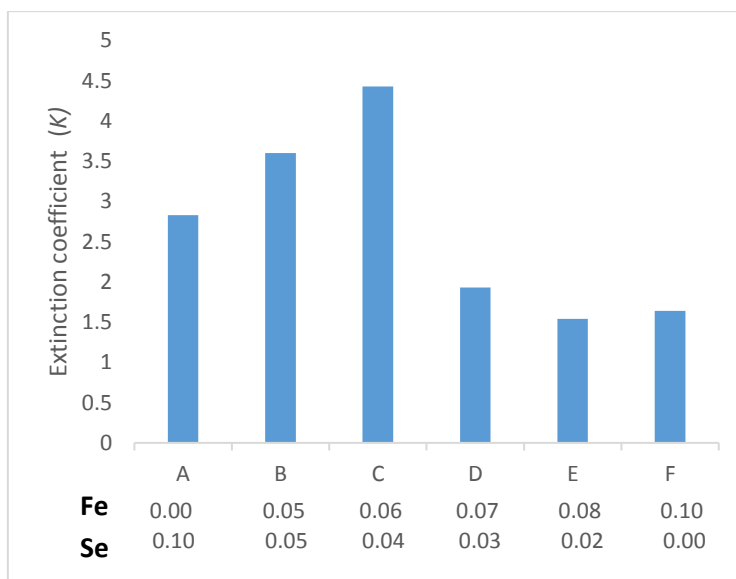


Figure 8, Extinction coefficient (K) versus sample compositions

V. Conclusion

Generally, the optical properties of thin-film ; transmission T , reflection R and absorption A are determined by its refractive index n , extinction coefficient K , bandgap E_g and geometry . Geometry include ; film thickness , thickness uniformity and film surface roughness. T , R and A are intrinsic - depending on the chemical composition and solid structure of the material, whereas the geometry is extrinsic[19].

In this work for the first time, the optical behaviour of Sn doped with varying molar concentrations of Fe:Semulticomposition thin films were studied. The Ternary thin films were deposited on soda lime glass by spray pyrolysis at a temperature of 400K . Spectroscopic measurements of the thin- films were carried out in the UV-Vis-NIR range at ambient temperature. The energy bandgaps exhibited by the film investigated fall within the energy range required for a semiconductor to effectively function as either buffer or window layers for photovoltaic applications. The small values of absorbance and reflectance in the visible region depicts that the the SnFeSe semiconductor materials could be useful as light emitting diodes (LED) and visible laser diodes (VLD).

Moreover, much is still needed to be done by researchers on the structural and electrical properties of this multicomposition thin films to compliment this research findings for industrial applications.

Acknowledgement

The authors would like to dedicate this work to Late Dr M.B.O Shitta who started the studies on FeSnSe –alloys but couldn't conclude it. We thank the entire Staff, Physics Laboratory unit, Sheda Science and Technology Complex (SHESTCO), Abuja-Nigeria for the assistance provided in our investigations.

References

- [1]. K. Annar, W.T Tan, N.Saravanan S.M.Ho and S.Y Gwce. Influence of PH values on chemical Bath deposited FeS₂ thin films *The pacific journal of science and technology* 10(2) November 2009
- [2]. D.Kathirvel, N. Suriyanaraganam, S. Prabahar S. Srikanth, P. Rajasekaram Structural, optical and electrical properties of chemical Bath deposited CdS thin film. *Chalcocgnide letter* 8(12) (2001) 739 -745
- [3]. P. Usha Rajalakashim, R.Oomen, C. Sanjeviraja. Strutral and optical characterization of bio>5 S₃ thin film *Chalcocgnide letter* 8(8) August, 2011 ,469-475.
- [4]. N.A. Okereke, A.J. Ekpunobi. Structure and optical properties of chemically deposited tin selenide .*Chalcocgnide letters* 7(9) ,2010 ,531-538.
- [5]. S.M.U Ishiwu. M.N Nnabuchi. The optical and solid state properties of lead selenide (PbSe) thin films grown by chemical Bath deposition technique. *Journal of Ovonic Research*, 6(2) 2010, 81-86.
- [6]. N.A. Okereke, A.J. Ekpunobi .Structural, optical properties and application of chemically deposited lead selenide thin films. *Journal of Ovonic Research* 6(6) Dec., 2010, 277-283.
- [7]. B.B. Nariya, A.K. Dasadia M.K. Bhayani A. J. Patel, A.R. Jani. Electric transport properties of SnS and SnSe simple crystals grown by direct vapour transport technique *chalcocgnide letters* 6(10) Oct., 2009, 549-554.
- [8]. D.D.O. Eya. Optical properties and applications of cadmium selenide (cdse) thin films prepared by chemical bath deposition techniques. *The pacific journal of science and technology*. 7(1) May, 2006 ,64-68.
- [9]. I.A. Ezenwa, N.A Okereke, N.S. Umeokwonna .Effect of PH on the optical properties of cadmium selenide (CdSe) thin films. *Journal of Basic Physical Research*, 2010,1(1) 9-12.
- [10]. M.C. Santhosh Kumar and B. Pradeep Electrical Properties of Silver selenide thin films prepared by reactive evaporation. *Bulletin of material science* 25(5), (2002) , 407-411.
- [11]. Y. M shakir, siddharta, G. Bhagavannaragana, M.A Wahab. Chemical bath deposition and formation of now materials by post deposition thermal processing superfice Structural, optical and electirical properties of ZnSe semiconductor nanoparticles *chalcocgnide letter* 8(7), July 2011, 455-440
- [12]. MTS Nair and P .K Nair, Antimony chalcocgnical thin films *Thin Solid Films* Volume 515, Issue 15, 31 May 2007, Pages 5777-5782
- [13]. M. Ramachandran, D. Martinez- ESCObar, J. S Narios- Rios, A. Sanchez- Jurez Lower plasma treatment for the groth of SnS thin film for photovoltaic application *Chalcocgnide letter* 8(11) Nov. 2011 695-697
- [14]. M.B.O. Shitta A.I. Mukolu and A.I. Anjorin. 1998 .The electrical properties of the preudo-binary alloy Fe_{1-x}Sn_xSe. (Unpublished. BSc thesis)
- [15]. D. B. Khadka and J. Kim .structural transition and bandgap tuning of Cu₂(Zn, Fe)SnS₄ chalcocgnides for photovoltaic application. *J. Phys. Chem.C*, 2014, 118 (26), pp 14227–14237
- [16]. I. Luminita, Duta A, Kriza A. Copper Sulphide obtained by Spray Pyrolysis, *Journal of Physics Conference Series* · April, 2007, 477-481.
- [17]. E. Lenei, I. Luminita, Duta A. Synthesis of alumic thin film by Spray Pyrolysis, 2010. N
- [18]. J.I Pankore. Optical process in Semiconductors, *Parentice-hall*, New York.
- [19]. I. J Cisneros. Optical characterization of dielectric and semiconductor thin films by use of transmission data .*Applied Optics*, 37, 5262, 1998,
- [20]. Optical nonlinearities in As₂Se₃ chalcocgnide glasses doped with Cu and Ag for pulse durations on the order of nanoseconds Kazuhiko Ogusu and Kenta Shinkawa *Opt. Express* 17(10) 8165-8172 (2009)
- [21]. Highly nonlinear As–S–Se glasses for all-optical switching
- [22]. J. M. Harbold, F. Ö. Ilday, F. W. Wise, J. S. Sanghera, V. Q. Nguyen, L. B. Shaw, and I. D. Aggarwal *Opt. Lett.* 27(2) 119-121 (2002)
- [23]. Structure, nonlinear properties, and photosensitivity of (GeSe₂)_{100-x}(Sb₂Se₃)_x glasses
- [24]. M. Olivier, J.C. Tchahame, P. Němec, M. Chauvet, V. Besse, C. Cassagne, G. Boudebs, G. Renversez, R. Boidin, E. Baudet, and V. Nazabal *Opt. Mater. Express* 4(3) 525-540 (2014)
- [25]. Improved nonlinear optical properties of chalcocgnide glasses in Ge-Sn-Se ternary system by thermal treatment

- [26]. Yicong Huang, Feifei Chen, Beijing Qiao, Shixun Dai, QiuhuaNie, and Xianghua Zhang
Opt. Mater. Express 6(5) 1644-1652 (2016)
- [27]. Pulse-width dependence of optical nonlinearities in As₂Se₃ chalcogenide glass in the picosecond-to-nanosecond region
- [28]. Kenta Shinkawa and Kazuhiko Ogusu Opt. Express 16(22) 18230-18240 (2008)

K. A. Aduloju "Tuning Some Optical Parameters Of $Se_{1-x}(A=Fe:Sn)$ Chalcogenides For photovoltaic Applications." IOSR Journal of Applied Physics (IOSR-JAP) , vol. 10, no. 1, 2018, pp. 48-57.